Magnetic Structures and Properties of Cr_{1-t}Fe_tAs

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The pseudo-binary CrAs-FeAs system has been investigated by X-ray and neutron diffraction and magnetic susceptibility measurements. $\text{Cr}_{1-t}\text{Fe}_t\text{As}$ is characterized by complete solid solubility, the structure of the ternary, random solid solution phase being of the MnP type. Cr-rich samples undergo a transition from MnP to NiAs type structure at high temperatures. The paramagnetic susceptibilities do not follow the Curie-Weiss Law. A double, c axis helimagnetic ordering is observed at low temperatures for $\text{Cr}_{1-t}\text{Fe}_t\text{As}$ samples with $0 \le t < 0.05$ and $\sim 0.82 < t \le 1.00$; for $0.10 \le t \le 0.80$ the samples exhibit weak ferromagnetism.

CrAs¹⁻³ and FeAs⁴ must be regarded as magnetically isostructural compounds, despite the fact that some of the parameters specifying their double, c axis spiral structures differ appreciably. Hence, the present study was started in order to establish the variations in the spiral parameters as function of the compositional parameter t for $Cr_{1-t}Fe_tAs$. However, even before this report was written, we have obtained data for similar quasi-binary systems (exemplified by the findings for $FeP_{1-x}As_x$ where a similar situation prevails for the binary end members⁴⁻⁶), which led us to question our original working hypothesis. A brief account of the magnetic and structural properties of $Cr_{1-t}Fe_tAs$ has previously been published by Kazama and Watanabe.⁷

EXPERIMENTAL

Samples of CrAs and FeAs were made from the elements [99.0 % Cr (powder crystals; Koch-Light Laboratories), 99.99 % Fe (turnings from rods; Johnson, Matthey & Co.), and 99.9999 % As (Koch-Light Laboratories)] according to the specifications in Refs. 2 and 4. Ternary $Cr_{1-t}Fe_tAs$ samples

were prepared by heating appropriate mixtures of CrAs and FeAs in evacuated and sealed quartz tubes at 850 °C for four periods of 8 days, interrupted by quenchings and intermediate crushings. Finally, the samples were cooled to room temperature over 3 days.

Details of the experimental methods (X-ray and neutron diffraction, magnetic susceptibility, and magnetization measurements) have been presented in previous communications.^{2,4} [The nuclear scattering lengths $b_{\rm Cr}=0.353$, $b_{\rm Fe}=0.951$, and $b_{\rm As}=0.64$ (in 10^{-12} cm) were taken from the 1976-compilation of The Neutron Diffraction Commission.]

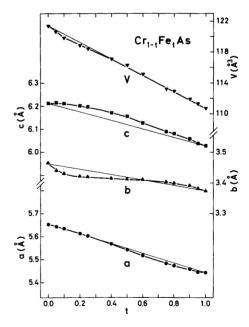


Fig. 1. Room temperature unit cell dimensions of ternary solid solution series CrAs—FeAs as functions of composition.

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Table 1. Unit cell dimensions and positional parameters with standard deviations for some $Cr_{1-t}Fe_tAs$ samples as derived by least squares profile refinements of neutron diffraction data. (Space group *Pnma*; positions 4(c); overall profile reliability factors ranging between 0.03 and 0.08.)

t	T(K)	a (Å)	b (Å)	c (Å)	X_T	z_T	x_X	z_X
0.05	10	5.6300(7)	3.3820(3)	6.2077(7)	0.0073(28)	0.1985(21)	0.1969(17)	0.5730(15)
	293	5.6403(5)	3.4350(2)	6.2095(4)	0.0075(20)	0.2007(16)	0.1995(12)	0.5753(11)
0.25	10	5.5929(6)	3.3840(2)	6.1814(5)	0.0048(9)	0.1974(7)	0.1997(7)	0.5752(7)
	293	5.5995(6)	3.4148(2)	6.1916(6)	0.0045(8)	0.2005(7)	0.2002(5)	0.5765(6)
0.50	10	5.5287(7)	3.3899(3)	6.1320(7)	0.0048(15)	0.1992(13)	0.1951(15)	0.5764(16)
	293	5.5386(9)	3.4115(5)	6.1418(7)	0.0048(18)	0.1995(13)	0.1972(15)	0.5769(16)
0.75	10	5.4772(16)	3.3786(6)	6.0852(12)	0.0061(10)	0.1980(8)	0.1964(13)	0.5748(13)
	293	5.4848(14)	3.4045(5)	6.0873(12)	0.0051(14)	0.1974(8)	0.1969(11)	0.5769(14)
0.90	10	5.4545(5)	3.3508(2)	6.0421(5)	0.0035(14)	0.1968(9)	0.1967(14)	0.5768(16)
	293	5.4700(5)	3.3932(3)	6.0578(5)	0.0060(19)	0.1998(13)	0.1999(19)	0.5760(18)
0.95	10	5.4569(5)	3.3455(2)	6.0386(5)	0.0052(16)	0.1966(9)	0.1990(15)	0.5764(17)
	293	5.4435(8)	3.3797(3)	6.0439(7)	-0.0001(19)	0.1916(11)	0.2001(19)	0.5841(20)

RESULTS AND DISCUSSION

(i) Chemical crystal structure. The continuous variations in the orthorhombic unit cell dimensions of Cr_{1-t}Fe_tAs with the compositional parameter t (Fig. 1) show that CrAs and FeAs exhibit complete, mutual solid solubility. A possible extension of the homogeneity range of Cr_{1-t}Fe_tAs to metal/nonmetal (atomic) ratios different from 1.00 has not been examined for $t \neq 0$ and 1. The X-ray and neutron diffraction data confirm that $Cr_{1-t}Fe_tAs$ $(0 \le t \le 1)$ takes the MnP type structure with random distribution of the two kinds of metal atoms at and below room temperature. Like CrAs the Cr-rich Cr_{1-t}Fe_tAs samples undergo a transition from MnP to NiAs type structure at temperatures of 1173 ± 20 ,8 1123 ± 50 , 1273 ± 50 , and 1373 ± 50 K for t = 0, 0.05, 0.15, and 0.30, respectively.

The unit cell dimensions and positional parameters (derived from neutron diffraction data by least squares profile refinements) at and below room temperature for samples with t=0.05, 0.25, 0.50, 0.75, 0.90, and 0.95 are listed in Table 1, values for t=0 and 1 being given in Refs. 2 and 4. Within the accuracy of the study, continuous changes in all variables with the composition are observed. This finding concurs with our observations on several isostructural, ternary phases. $^{6,9-15}$ The magnetic orderings in $Cr_{1-t}Fe_tAs$ take place below room temperature (see iii) and their occurrence has

accordingly no influence on the room temperature unit cell dimensions. The marked cusps at t=0.5 in the curves for a and c, reported by Kazama and Watanabe, 7 are not verified in this study.

(ii) Magnetic susceptibility. The thermomagnetic curves for some fifteen samples covering the entire homogeneity range of Cr_{1-t}Fe_tAs were registered between 4.2 and 1000 K. There is a systematic trend in the level and shape of the $\chi^{-1}(T)$ curves with the compositional parameter t, and, for the purpose of clarity, Fig. 2 includes only the results for about half of the samples which were measured. In the range $0.10 \le t \le 0.80$ a ferromagnetic behaviour of the samples was observed at low temperatures. As evident from Fig. 2, the Curie temperature goes through a maximum value in the vicinity of t = 0.5, where also a maximum in magnetization (at 4.2 K) is observed. As an example of this behaviour, the complete thermomagnetic curve for $Cr_{0.50}Fe_{0.50}As$ is presented in Fig. 3.

The present findings for $\mathrm{Cr_{1-r}} Fe_{\mathrm{r}} As$ are essentially in good agreement with the data published by Kazama and Watanabe. Within the limiting accuracies of the two studies there is a perfect match in the Curie temperatures. The paramagnetic moments ($\mu_{\mathrm{P}} = \sqrt{8C_{\mathrm{mol}}}$) as derived from the slopes of the $\chi^{-1}(T)$ curves at 300 K, show little variation over the interval $0.20 \leq t \leq 0.80$ (changing from $2.4 \pm 0.2~\mu_{\mathrm{B}}$ for t = 0.20 to $2.1 \pm 0.2~\mu_{\mathrm{B}}$ for t = 0.80), concurring roughly with the results of Kazama and

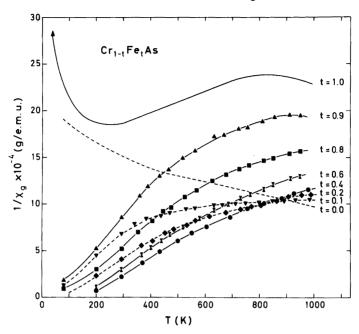


Fig. 2. Inverse magnetic suceptibility versus temperature for some $Cr_{1-t}Fe_tAs$ samples. Results for CrAs and FeAs are quoted from Refs. 11 and 4.

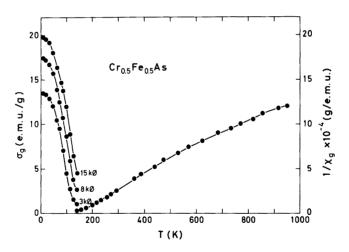


Fig. 3. Thermomagnetic data for Cr_{0.50}Fe_{0.50}As.

Watanabe. However, as noted for the unit cell dimensions, the present data show no cusp in μ_P at t=0.5. It should nevertheless be emphasized that the evaluation of the magnetic moments from the present data are rather dubious, since, actually,

none of our $\chi^{-1}(T)$ curves (cf., Figs. 2 and 3) follow the Curie-Weiss Law.

(iii) Magnetic structures. Both CrAs² and FeAs⁴ take a helimagnetic arrangement of the double, c axis type below 261-272 and 77 ± 1 K, respectively

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t	T(K)	$\tau/2\pi c^*$	$\mu_{ m H}(\mu_{ m B})$	φ (°)	$T_{N}(K)$
1.00	12	0.375 + 0.005	0.5 + 0.1	140+10	77 + 1
0.95	10	0.452 ± 0.005	0.7 + 0.1	141 + 10	56 + 2
0.90	10	0.456 + 0.005	0.6 ± 0.1	137 + 10	37 ± 3
0.00	80	0.353 ± 0.003	1.70 ± 0.05	-133 + 1	261 - 272

Table 2. Helimagnetic parameters and Néel temperatures for Cr₁₋₁Fe_tAs. (Angle between magnetic moment and propagation vector fixed at 90°.)

(Table 2). The neutron diffraction diagram of $Cr_{0.95}Fe_{0.05}As$ at 10 K shows neither purely antiferromagnetic nor helimagnetic reflections. The sample does not undergo the characteristic first order phase transition (see Table 1) from the paramagnetic to the cooperative magnetic state as found for $CrAs^2$ and CrAs-rich samples of $V_{1-t}Cr_tAs$, $Cr_{1-t}Mn_tAs$, $Cr_{1-t}Co_tAs$, and $CrP_{1-x}As_x$. The helimagnetic ordering in CrAs extends only slightly (t < 0.05) into the ternary range of $Cr_{1-t}Fe_tAs$.

The helimagnetic ordering in FeAs, however, extends to $t = \sim 0.82$. No pronounced changes are observed in the parameters specifying the spiral structure [i.e., the magnetic moment ($\mu_{\rm H}$), the propagation vector (τ) and the phase angle between independent spirals (ϕ)] on going from FeAs to ${\rm Cr_{0.10}Fe_{0.90}As}$ (Table 2). It may, however, be interesting to note the increase in τ from $0.375 \times 2\pi c^*$ in FeAs to $0.456 \times 2\pi c^*$ in ${\rm Cr_{0.10}Fe_{0.90}As}$, the latter being the largest propagation vector observed for helimagnetic MnP type phases.

The substitution of FeAs by CrAs leads to an appreciable lowering of the Néel temperature (T_N) from 77 ± 1 K for FeAs to 56 ± 2 K for t=0.95 and 37 ± 3 K for t=0.90 (average values determined from the temperature dependences of the intensities of 101^- and 000^\pm , cf. Fig. 4). The uncertainties in T_N also reflect the problem of estimating the background level in the neutron diffraction diagrams. The almost linear dependence between T_N and t suggest a magnetic phase boundary at $t=\sim0.82$ (where T_N extrapolates to 0 K). We have not found it worthwhile to investigate samples with $\sim0.82 < t < 0.90$.

The slightly different shapes of the intensity curves (integrated intensity versus temperature) for the satellite reflections 000^{\pm} and 101^{-} (Fig. 4) indicate minor variations in ϕ with the temperature. The propagation vector increases slightly with increasing temperature. Due to the few magnetic reflections and their low intensities the magnetic parameters

for $Cr_{0.05}Fe_{0.95}As$ and $Cr_{0.10}Fe_{0.90}As$ are encumbered with relatively large uncertainties (cf., FeAs⁴). The helimagnetic ordering of FeAs extends further into the ternary region of $Cr_{1-t}Fe_tAs$ than has been observed for $Mn_{1-t}Fe_tAs$, $Fe_{1-t}Co_tAs$, and $FeP_{1-x}As_x$.

The rather weak ferromagnetism of $Cr_{1-t}Fe_tAs$ (0.10 $\leq t \leq$ 0.80) observed by magnetic susceptibility and magnetization measurements, is difficult to explore by the powder neutron diffraction technique. A ferromagnetic model based upon a magnetic moment along b (viz., in analogy with the ferromagnetic mode observed for MnP ¹⁶) gave only a minor improvement in the nuclear and profile R factors and a rather unsatisfactory value for the magnetic

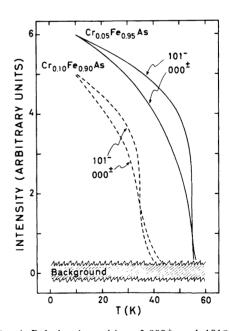


Fig. 4. Relative intensities of 000^{\pm} and 101^{-} as functions of temperature for $Cr_{0.05}Fe_{0.95}As$ (solid lines) and $Cr_{0.10}Fe_{0.90}As$ (broken lines).

R factor for the sample $\mathrm{Cr_{0.50}Fe_{0.50}As}$. The best fit was, however, obtained with a magnetic moment of 0.5 μ_{B} at 10 K, which is in accordance with the value from magnetization measurements. The corresponding refinements for $\mathrm{Cr_{0.75}Fe_{0.25}As}$ and $\mathrm{Cr_{0.25}Fe_{0.75}As}$ were not successful, due to a still smaller ferromagnetic moment in these samples. Thus, there is an appreciable reduction in the magnetic moment of $\mathrm{Cr_{1-r}Fe_rAs}$ on going from the paramagnetic to the ferromagnetic state.

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