# On the Crystal Structures of FeOHSO, and InOHSO,

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The basic iron sulfate, FeOHSO<sub>4</sub>, forms orthorhombic crystals with lattice parameters a=7.33, b=6.42, and c=7.14 Å. The space group is  $D_{2k}^{16}-Pnma$  and the unit cell contains four formula weights. Each iron atom is octahedrally coordinated with two hydroxogroups and four sulfate oxygens. Each hydroxogroup is bonded to two iron atoms. The structure of the basic indium sulfate, InOHSO<sub>4</sub>, has been shown to be similar.

Crystal structure determinations of a number of basic salts of group 3b metals have shown that the metal atoms in these salts are linked together by means of single or double bridges of hydroxogroups. In crystals of  $AlOHSO_4(H_2O)_5$  the aluminum atoms are joined by double OH bridges, which results in the formation of discrete aluminum oxygen complexes <sup>1</sup>. In the corresponding basic sulfates of indium and thallium,  $InOHSO_4(H_2O)_2^2$  and  $TlOHSO_4(H_2O)_{2.5}^3$  the metal atoms are joined by single OH bridges leading to infinite metal-oxygen chains. These basic salts can all be obtained from solutions at room temperature. To see if the same type of bridging occurs in basic salts prepared under different conditions, the structure of an indium sulfate formed by hydrothermal hydrolysis at high temperature has been investigated. It has the composition  $In_2O_3 \cdot 2SO_3 \cdot H_2O$ . An iron sulfate with the same composition,  $Fe_2O_3 \cdot 2SO_3 \cdot H_2O$ , can be obtained by the same method of preparation <sup>4</sup>. Structurally the two compounds are closely related.

#### **EXPERIMENTAL**

Preparation of crystals.  $\rm In_2O_3\cdot 2SO_3\cdot H_2O$ : Solutions of indium sulfate with concentrations ranging from 0.004 to 1.0 M were prepared by dissolving  $\rm In_2(SO_4)_3$  in distilled water or in weak sulfuric acid ( < 15 %  $\rm H_2SO_4$ ). Small samples of the solutions were then heated in sealed pyrex tubes at 100°, 160°, or 190°C for about a week. No precipitates were formed in the tubes heated to 100°C. At the higher temperatures two phases were obtained. One formed only in the most dilute solutions and did not contain sulfate. It was shown by powder photographs to be  $\rm In(OH)_3$ . The other was the basic sulfate  $\rm In_2O_3\cdot 2SO_3\cdot H_2O$ . It crystallized as thin needles which were bound together into larger aggregates of rather irregular shapes with the needle axes parallel.

Fe<sub>2</sub>O<sub>3</sub>·2SO<sub>3</sub>·H<sub>2</sub>O: The range of stability has been given by Posjnak and Merwin <sup>4</sup>. Crystals sufficiently large for X-ray work, were obtained by heating a solution containing 4.7 % Fe<sub>2</sub>O<sub>3</sub>, 12.9 % SO<sub>3</sub>, and 82.4 % H<sub>2</sub>O at 200°C for about a month.

Analysis. The results of the analysis are compared in Table 1 with calculated values

for In<sub>2</sub>O<sub>3</sub>·2SO<sub>3</sub>·H<sub>2</sub>O and Fe<sub>2</sub>O<sub>3</sub>·2SO<sub>3</sub>·H<sub>2</sub>O.

			Table 1.			
	$In_2O_3 \cdot 2SO_3 \cdot H_2O$			$Fe_2O_3 \cdot 2SO_3 \cdot H_2O$		
	Found	Calc.		$\mathbf{Found}$	Calc.	
$\% \text{ In}_{2}O_{3}$	60.7	60.91	% Fe <sub>2</sub> O <sub>3</sub>	47.2	47.27	
% SO <sub>3</sub>	35.1	35.13	% SO <sub>3</sub>	47.0	47.40	
% H <sub>2</sub> O	4.5	3.95	% H <sub>2</sub> O	5.9	5.33	
Density	$\mathbf{3.9_4}$	4.05	Density	3.2,	3.34	

The amount of water was determined according to the method of Brush and Penfield 5, and sulfate was determined as BaSO<sub>4</sub>. In<sup>3+</sup> and Fe<sup>3+</sup> were precipitated as hydroxides. After ignition the residue was weighed as In<sub>2</sub>O<sub>3</sub> or Fe<sub>2</sub>O<sub>3</sub>. The density was obtained by comparison of the weight of a sample in benzene and in air.

 $\hat{X}$ -Ray data. Rotation and Weissenberg photographs taken around the three crystallographic axes showed that the crystals are orthorhombic. The following values for the unit cell dimensions were calculated from powder photographs taken with CuKa radiation ( $\lambda=1.54050$  Å) in a Guinier focusing camera with KCl ( $a_0=6.2929$  Å) as internal standard:

(Estimated accuracy  $\pm 0.005$  Å). The first lines on the powder photographs are given in Table 2.

A comparison between observed and calculated densities shows that there are two

formula weights, thus 4 Fe(In) atoms, in the unit cell (Table 1).

For Fe<sub>2</sub>O<sub>3</sub>·2SO<sub>3</sub>·H<sub>2</sub>O the systematically absent reflections are 0kl with k+l odd and hk0 with h odd. This is characteristic for the space groups No. 62: Pnma and No. 33:  $Pn2_1a$ . The non-centrosymmetric  $Pn2_1a$  was not considered as it was possible to find a satisfactory structure with the higher symmetry.

The crystals are extended along the a axis. The odd layer lines of rotation photographs taken around this axis are very weak. On the even layer lines all reflections with  $\frac{1}{2}h + k + l$  odd are absent, which indicates that for every atom at x, y, z there must also be an atom at  $\frac{1}{4} + x$ ,  $\frac{1}{2} + y$ ,  $\frac{1}{2} + z$  or  $\frac{3}{4} + x$ ,  $\frac{1}{2} + y$ ,  $\frac{1}{2} + z$ .

Difficulties were encountered in finding good single crystals of the corresponding indium compound. The spots on the Weissenberg films were often rather extended and difficulties.

diffuse. This was especially true for crystals prepared at 190°C. Weissenberg photographs showed that the systematic absences are the same as for Fe<sub>2</sub>O<sub>2</sub>·2SO<sub>2</sub>·H<sub>2</sub>O except for the presence of very weak reflections hk0 for h odd. The odd layer lines around the a axis are extremely weak in comparison with the even layer lines which have the same systematic absences for  $\frac{1}{2}h + k + l$  odd as was noted for the Fe salt.

For the collection of intensity data Weissenberg photographs were taken around all three axes using the multiple film technique with MoK radiation for the Fe salt and  $\mathrm{Cu}K$  radiation for the In salt. The maximum dimension of the crystals used were  $0.2~\mathrm{mm}$ for the Fe and 0.1 mm for the In salts. The intensities were estimated visually by comparison with a calibrated intensity strip and were corrected for Lorentz and polarization

Because Fe<sub>2</sub>O<sub>3</sub>·2SO<sub>3</sub>·H<sub>2</sub>O has a higher symmetry and gave more reliable intensity data, a complete structure determination was carried out only for this compound.

### STRUCTURE DETERMINATION

The very weak odd layer lines around the a axis for both the iron and the indium compound indicate that there is no or at least only a very small contri-

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bution to these reflections from the metal atoms. Thus, if one Fe atom has the position xyz, there must be a corresponding atom at  $\frac{1}{2} + x, y, z$ . The extinction on the even layer lines of reflections with  $\frac{1}{2}h + k + l$  odd further shows that there must also be Fe atoms at 1/4 + x,  $\frac{1}{2} + y$ ,  $\frac{1}{2} + z$  and 3/4 + x,  $\frac{1}{2} + y$ ,  $\frac{1}{2} + z$ . The only position in space group Pnma that can accomodate four Fe atoms related in this way, is 4(c):  $\pm(x, 1/4, z)$ ,  $\pm(\frac{1}{2} + x, 1/4, \frac{1}{2} - z)$ . The parameter values must then be  $x = \pm 1/8$  or  $\pm 3/8$  and  $z = \pm 1/4$ . All these positions are equivalent and any set of parameters can be chosen, for example: x = 1/8, z = 1/4, (y = 1/4).

The Fe atoms make a maximum contribution to all 0kl reflections and therefore probably determine their signs. On this assumption the electron density projected on the yz plane was calculated. As no signs were later changed because of the contributions from the light atoms, the resulting Fourier map was the same as the final electron density map shown in Fig. 1a. It reveals clearly the positions of the four sulfate groups of the unit cell, and it also indicates the positions of the four oxygen atoms  $(O_4)$  not associated with the sulfate groups.

According to Fig. 1a the parameters of  $O_4$  are y=1/4 and  $z\approx 0.15$ , and  $O_4$  therefore occupies the position 4(c). The two nearest Fe atoms are at 1/8, 1/4, 1/4 and 5/8, 1/4, 1/4. The  $O_4$  atom must have a position between these Fe atoms, that is with x=3/8 (or 7/8 which is structurally equivalent), as other values of x would result in Fe-O<sub>4</sub> distances, which are too short. O<sub>4</sub> is therefore in contact with two Fe atoms at distances of approximately 2.0 Å.

The possible x parameters for the sulfur atoms are restricted in the same way as the parameters for the Fe atoms, that is they must be  $x = \pm 1/8$  or  $\pm 3/8$ . With the position of O<sub>4</sub> given  $(x = 7/8, y = 1/4, z \approx 0.35)$  and with

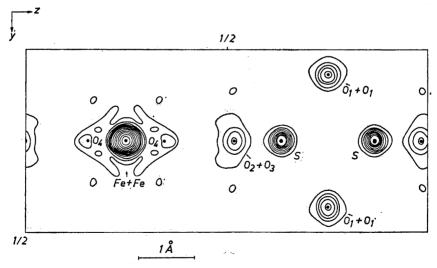


Fig. 1 a. Electron density projection along the a axis for FeOHSO<sub>4</sub>. Contours are drawn at intervals of  $10 \, \mathrm{e} \cdot \text{Å}^{-2}$ . The zero contour is omitted. At the Fe peak only every 3rd contour has been drawn above  $50 \, \mathrm{e} \cdot \text{Å}^{-2}$ . The final atomic positions are indicated by dots.

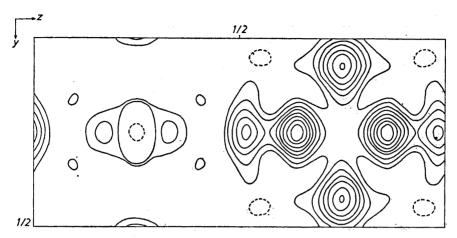


Fig. 1 b. Electron density projection along the a axis for InOHSO<sub>4</sub> with the contribution from the In atoms subtracted. Contours are drawn at intervals of  $4 e \cdot Å^{-2}$ . The zero contour is omitted. Broken lines indicate negative regions.

the sulfate group oriented as shown by the 0kl projection in Fig. 1a, the only possible parameters for the fourfold sulfur position, not leading to O-O distances, which are too short, are x=3/8 and y=1/4,  $z\approx0.63$ .

Of the oxygen atoms of the sulfate groups,  $O_1$  occupies an eightfold position since according to Fig. 1a it lies outside the mirror planes at  $y=\pm 1/4$ .  $O_2$  and  $O_3$  lie on the mirror planes, that is they are in fourfold positions. Their x

Table 2. Powder photographs ( $\sin^2\Theta < 0.20$ ) of FeOHSO<sub>4</sub> and InOHSO<sub>4</sub>, taken with CuKa radiation in a Guinier camera.

		HSO <sub>4</sub>				HSO <sub>4</sub>	
	$\sin^2\!\Theta$	$\sin^2\!\Theta$	I		$\sin^2\Theta$	$\sin^2\!m{\Theta}$	I
$h \ k \ l$	calc.	obs.	obs.	$h \ k \ l$	calc.	obs.	obs.
101	0.0227	0.0226	vvw	101	0.0211	0.0211	vvw
011	0.0260	0.0260	$\mathbf{m}$	011	0.0245	0.0245	$\mathbf{w}$ +
111	0.0371	0.0370	vvw	<b>002</b>	0.0443	0.0443	$\mathbf{w}$ +
002	0.0465	0.0465	w	201	0.0514	0.0514	vs
201	0.0558	0.0558	vs	020	0.0537)		
102	0.0575)			210	0.0537	0.0537	s
020	0.0576	0.0576	s	102	0.0543	0.0545	vvw
112	0.0719	0.0719	vw	$\boldsymbol{022}$	0.0979		
212	0.1050	0.1050	vvw	212	0.0980	0.0980	$\mathbf{m}$
<b>221</b>	0.1134	0.1134	w	221	0.1051	0.1052	s
013	0.1190	0.1190	W	013	0.1130	0.1131	w
031	0.1413	0.1413	vw	031	0.1318	0.1318	vw
203	0.1488	0.1488	w	203	0.1399	0.1399	$\mathbf{w} +$
230	0.1738	0.1741	vvw	230	0.1611)	0.1612	
400	0.1766	0.1765	w	400	0.1613	0.1012	$\mathbf{m}$
004	0.1860	0.1859	vvw	004	0.1770'	0.1775	vvw
				411	0.1858	0.1857	w
				223	0.1936	0.1935	vw

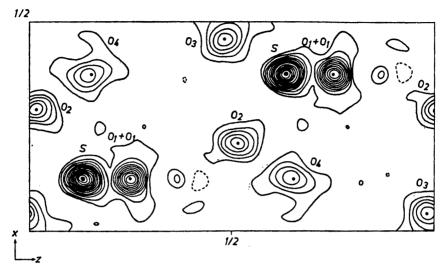


Fig. 2. Electron density projection along the b axis for FeOHSO<sub>4</sub> with the contribution from the Fe atoms subtracted. Contours are drawn at intervals of b e·b-2. The zero contour is omitted. The final atomic positions are indicated by dots.

parameters can be calculated assuming the O—O distance in a sulfate group to be 2.4 Å.

The resulting parameter values were refined from difference syntheses and electron density projections along the a, b, and c axes. The refinement was continued until the difference maps did not indicate any significant shifts in the positional parameters. The final parameters are given in Table 3. The final electron density projection along the b axis with the contribution from the Fe atoms subtracted is shown in Fig. 2.

Of the atoms of the unit cell only  $O_2$  and  $O_3$  give contributions to reflections for which h is even and  $\frac{1}{2}h + k + l$  is odd. To maintain the condition that these reflections are systematically absent the parameters of  $O_2$  and  $O_3$  must be related in such a way, that, for example, an  $O_2$  atom at xyz has a corresponding  $O_3$  atom at 1/4 + x (or 3/4 + x),  $\frac{1}{2} + y$ ,  $\frac{1}{2} + z$ . The final parameters in Table 3 are in accordance with this as there are no indications in the difference maps of any significant deviations from this relation.

Table 3.

Final	parameters
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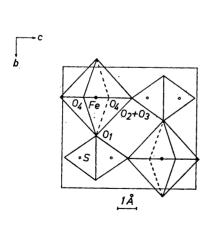
	$oldsymbol{x}$	$oldsymbol{y}$	$\boldsymbol{z}$
4 Fe in $4(c)$	0.125	1 <b>/4</b>	0.250
4 S in 4(c)	0.875	1/4	0.866
$8 O_1 \text{ in } 8(d)$	0.375	0.067	0.750
$4 O_{2} in 4(c)$	0.209	1/4	0.517
$4 O_{3} \text{ in } 4(c)$	0.959	3/4	0.017
4 $O_4$ in $4(c)$	0.375	1/4	0.155

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For the final calculation of structure factors listed in Table 5, temperature factors of 0.3 Å<sup>2</sup> for Fe and S and 0.5 Å<sup>2</sup> for O atoms were used. The scattering factors were those given by Berghuis et al.6 for O, by Tomiie and Stam 7 for S, and by Thomas and Umeda 8 for Fe. The R factors  $(R = \Sigma / |F_o| - |F_c| / |\Sigma| F_o|)$ with only observed reflections included are 0.084 for 0kl, 0.070 for hol, and 0.086 for hk0 reflections. For reflections with h odd there are no contributions from the Fe or  $O_1$  atoms. For this class of reflections the R factors are 0.11 for hol and 0.13 for 1kl reflections. Standard deviations in positional parameters estimated according to the formula given by Cruickshank 9 are about 0.005 Å for S, 0.01<sub>5</sub> Å for O<sub>1</sub>, O<sub>2</sub>, and O<sub>3</sub>, and 0.02 Å for O<sub>4</sub>. Since most of the atoms are in special positions, the standard deviations in bond lengths will, however, generally be smaller than might be expected from the values given. The structure of In<sub>2</sub>O<sub>3</sub>·2SO<sub>3</sub>·H<sub>2</sub>O is essentially the same as that of Fe<sub>2</sub>O<sub>3</sub>·2SO<sub>3</sub>·H<sub>2</sub>O. The electron density projection calculated from the 0kl structure factors with the contributions from the indium atoms subtracted is shown in Fig. 1b. It is in all essentials identical with that of the iron compound (Fig. 1a), but the presence of weak reflections hk0 for h odd shows that the symmetry of the indium salt is lower. The differences probably can be explained by rather small deviations of the parameter values of the light atoms in In<sub>2</sub>O<sub>3</sub>·2SO<sub>3</sub>·H<sub>2</sub>O from the Pnma symmetry. However, attempts to determine these deviations by means of trial and error were not successful and failed to give an R factor below about 0.30 for the weak 1kl and 3kl reflections, compared with a R factor of about 0.10 for the 0kl and 2kl reflections.

## DISCUSSION OF THE STRUCTURE

Projections of the structure along the a and b axes are shown in Fig. 3. Each metal atom is octahedrally surrounded by six oxygen atoms. Two of



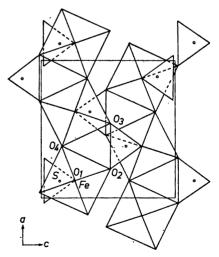


Fig. 3 a. Projection of the structure of FeOHSO<sub>4</sub> along the a axis.

Fig. 3 b. Projection of the structure of  $FeOHSO_4$  along the b axis.

Table 4. Interatomic distances.

Distances within a FeO, octahedron:

Distances within a SO<sub>4</sub> group:

Other O-O distances < 4.0 Å

these,  $O_4$ , are shared between two octahedra. The remaining four belong to sulfate groups. The four hydrogen atoms of the unit cell are probably associated with the bridging oxygens,  $O_4$ , as this would increase the sum of electrostatic bond strengths for these atoms from 1 to the expected value of 2. The metal atoms are therefore joined by single hydroxobridges, and the formulae should be written as  $FeOHSO_4$  and  $InOHSO_4$ . The interatomic distances (Table 4) give no evidence for the presence of hydrogen bonds.

The same kind of single hydroxobridges are present in the structure of  $InOHSO_4(H_2O)_2^2$ . The two structures are in fact closely related. The two water oxygens of the  $InO_6$  octahedra in  $InOHSO_4(H_2O)_2$  are replaced by sulfate oxygens in  $InOHSO_4$ . The infinite In-OH-In chains, which in the former compound are joined to other similar chains only by means of hydrogen bonds, are in  $InOHSO_4$  and  $FeOHSO_4$  joined to other chains by means of oxygen atoms belonging to the same sulfate group. The In-In distance, 3.84 Å, within an In-OH-In chain in  $InOHSO_4$  is slightly smaller than the corresponding distance in  $InOHSO_4(H_2O)_2$ , 3.95 Å. Accurate values for the In-O distances can not be calculated, but the approximate structure shows that they do not differ much from the values found for  $InOHSO_4(H_2O)_2^2$ .

In FeOHSO<sub>4</sub>, the FeO<sub>6</sub> octahedra are very nearly regular. In view of the standard deviations, however, the difference between the shortest of the Fe—O bond lengths (1.95 Å and involving the bridging oxygens, O<sub>4</sub>) and the longest (2.04 Å) is probably significant. The average value of the Fe—O distances, 2.00 Å, is close to the sum of the crystal radii, 2.04 Å <sup>10</sup>.

The average S—O distance of 1.46 Å does not differ significantly from S—O distances found for sulfate groups in other structures <sup>1</sup>.

Table~5. Observed and calculated structure factors for FeOHSO4.

hkl	$F_{\mathbf{o}}$	$F_{\mathbf{c}}$	hkl	$F_{ m o}$	$F_{ m c}$
000 2 4 6 8 10 12 14 16 18	$ \begin{array}{c}  - \\  102 \\  87 \\  39 \\  83 \\  52 \\  27 \\  < 17 \\  23 \\  21 \end{array} $	$   \begin{array}{r}     328 \\     -104 \\     83 \\     -37 \\     86 \\     -53 \\     27 \\     -17 \\     23 \\     -23 \\   \end{array} $	060 2 4 6 8 10 12 14 16 18	$ \begin{array}{c} 112 \\ 54 \\ 37 \\ 28 \\ 60 \\ 39 \\ 21 \\ < 17 \\ 20 \\ 14 \end{array} $	-116 $49$ $-37$ $23$ $-63$ $41$ $-22$ $14$ $-19$
011 3 5 7 9 11 13	$\begin{array}{c} 52 \\ 85 \\ 57 \\ 35 \\ < 11 \\ 41 \\ 26 \\ 21 \end{array}$	$\begin{array}{r} - & 46 \\ 91 \\ - & 50 \\ 35 \\ - & 3 \\ 41 \\ - & 26 \\ 20 \end{array}$	071 3 5 7 9 11 13	$\begin{array}{c} 30 \\ 65 \\ 50 \\ 31 \\ < 16 \\ 34 \\ 26 \\ 19 \end{array}$	$egin{array}{c} 30 \\ -58 \\ 44 \\ -30 \\ 8 \\ -35 \\ 23 \\ -19 \end{array}$
020 2 4 6 8 10	155 21 28 < 8 63 32	-176 $20$ $-26$ $3$ $-60$ $35$	080 2 4 6 8	$\begin{array}{c} 64 \\ 20 \\ < 14 \\ < 15 \\ 36 \\ 24 \end{array}$	$     \begin{array}{r}       65 \\       -17 \\       8 \\       -2 \\       37 \\       -21     \end{array} $
031 3 5 7 9 11 13 15	88 109 80 47 17 49 32 23	$egin{array}{c} 90 \\ -117 \\ 76 \\ -53 \\ 18 \\ -51 \\ 34 \\ -27 \\ \end{array}$	091 3 5 7 9 11 13	$\begin{array}{c} 29 \\ 52 \\ 46 \\ 26 \\ < 17 \\ 31 \\ 25 \end{array}$	$\begin{array}{r} - \ 27 \\ 50 \\ - \ 39 \\ 28 \\ - \ 10 \\ 33 \\ - \ 21 \end{array}$
040 2 4 6 8 10 12 14	131 41 32 16 60 36 < 17 < 17	$egin{array}{c} 140 \\ -39 \\ 32 \\ -13 \\ 62 \\ -37 \\ 17 \\ -9 \\ \end{array}$	0100 2 4 6 8 10	64 34 < 16 18 35 30	- 64 27 - 20 13 - 40 26
16 18 051 3 5 7	19 15 22 54 41	16 - 17 - 19 - 33 - 36	$\begin{array}{c} 3 \\ 5 \\ 0120 \\ 2 \end{array}$	$< 16 \\ 24 \\ 23 \\ 42 \\ 26 \\ -$	$ \begin{array}{r}     8 \\     -26 \\     19 \end{array} $ $ \begin{array}{r}     48 \\     -20 \end{array} $
$7 \\ 9 \\ 11 \\ 13$	$< egin{array}{c} 20 \ 15 \ 33 \ 22 \ \end{array}$	$\begin{array}{c} 22 \\ 2 \\ 32 \\ -19 \end{array}$	4 6 8 10	$< 17 \\ 19 \\ 29 \\ 21$	15 10 31 - 21

hkl	$F_{\mathrm{o}}$	$oldsymbol{F_c}$	hkl	$F_{ m o}$	$F_{ m c}$
0131 3 5 7 9 11	< 17 31 28 16 < 15 18	$ \begin{array}{rrr}  & -18 \\  & 30 \\  & -24 \\  & 19 \\  & -9 \\  & 22 \end{array} $	131 2 3 4 5 6 7 8	25 38 < 6 < 7 < 7 17	$\begin{array}{r} - & 24 \\ - & 39 \\ & 0 \\ & 7 \\ & 5 \\ & 17 \\ - & 15 \end{array}$
0140 2 4 6 8	$\begin{array}{c} 26 \\ < 17 \\ < 16 \\ < 16 \\ 16 \end{array}$	- 29 8 - 6 2 - 18	9 10	$     \begin{array}{r}       14 \\       < 11 \\       11     \end{array} $ $     \begin{array}{r}       21 \\       33 \\       < 7     \end{array} $	$ \begin{array}{rrr}  & -15 \\  & -9 \\  & -13 \end{array} $
0151 3 5 7	< 16 < 16 18 19	$-\begin{array}{c} 11 \\ -20 \\ 15 \\ -13 \end{array}$	141 2 3 4 5 6 7	< 8 < 8 13 10	$ \begin{array}{rrrr}  & -33 \\  & -2 \\  & 6 \\  & -5 \\  & 16 \\  & 15 \end{array} $
0160 2 4 6 8	$\begin{array}{c} 24 \\ < 14 \\ < 13 \\ < 12 \\ 16 \end{array}$	30 14 13 9 21	$\begin{matrix} 8\\9\\10\end{matrix}$	< 10 13 16	$ \begin{array}{rrr}  & -14 \\  & 9 \\  & -13 \\ \end{array} $
101 2 3 4 5 6 7 8	$\begin{array}{c} 34 \\ 53 \\ 9 \\ 10 \\ < \begin{array}{c} 7 \\ 19 \\ 17 \end{array}$	35 - 55 4 9 - 4 18 16	151 2 3 4 5 6 7 8	25 < 8 < 8 11 17 17 15	29 - 3 - 6 - 5 - 15 - 14
10 111	16 13 12 29	$-16 \\ 10 \\ -14$	161 2 3 4 5 6	14 22 < 9 < 9 < 9	$ \begin{array}{rrr}  & -15 \\  & 25 \\  & 4 \\  & -5 \\  & 5 \\  & -14 \end{array} $
2 3 4 5 6 7 8	$\begin{array}{c} 49 \\ 9 \\ 6 \\ 10 \\ 20 \\ 20 \\ 17 \end{array}$	52 $3$ $-$ 9 $-$ 5 $-$ 18 $16$ $16$	7 171 2 3 4	12 12 18 < 9 < 10 < 10	-13 $-14$ $-22$ $4$
121 2 3 4	$\begin{array}{c} 30 \\ 47 \\ < 6 \end{array}$	- 29 46 - 2 - 8	5 6 7	< 10 13 14	5 4 13 - 12
4 5 6 7 8 9	$< \begin{array}{c} 13 \\ 7 \\ 14 \\ 11 \\ 10 \end{array}$	$ \begin{array}{r}     5 \\     - 17 \\     - 16 \\     16 \end{array} $	$181 \\ 2 \\ 191 \\ 2$	18 < 11 15	$-\begin{array}{c} 12 \\ 20 \\ 11 \\ 18 \end{array}$
9 10 11	10 13 17 12	-	$^{1101}_{2}$	$< 12 \\ 14$	$-\   \begin{array}{c} 10 \\ 16 \end{array}$

hkl	$F_{o}$	$F_{\mathbf{c}}$	hkl	$F_{\mathbf{o}}$	$F_{\mathrm{c}}$
201 3 5	135 97 55	-149 95 - 59	1005 7 9	24 44 35	- 25 45 - 36
7 9 11	78 60 30	79 - 59 34	1101 2	< 15 19	- 10 - 16
301 2 3 4 5	22 39 36 21 24	$ \begin{array}{r}     18 \\     -34 \\     31 \\     19 \\     25 \end{array} $	1200 2 4 6 8	$\begin{array}{c} 56 \\ 27 \\ 24 \\ < 19 \\ 41 \end{array}$	$ \begin{array}{rrr}  & -65 \\  & 30 \\  & -23 \\  & 16 \\  & -41 \end{array} $
6 400 2 4 6 8	29 125 116 28 55 63 53	29	1401 3 5 7 1600 2	$egin{array}{c} 27 \\ < 20 \\ < 20 \\ 30 \\ \\ 26 \\ 23 \\ \\ \end{array}$	$ \begin{array}{r}     34 \\     -23 \\     18 \\     -30 \end{array} $ $ \begin{array}{r}     31 \\     -27 \end{array} $
501 2 3 4 5 6	19 33 < 9 < 9 < 10	- 19 32 2 - 6 5 - 16	210 3 5 7 9	16 69 < 9 17 16	- 16 64 - 4 19 - 18
7 8 9 10	<ul> <li>18</li> <li>15</li> <li>16</li> <li>17</li> </ul>	$-\begin{array}{c} -14 \\ 14 \\ -9 \\ 12 \end{array}$	420 4 6 8	79 87 75 46 49	$ \begin{array}{r}     77 \\     -79 \\     78 \\     -43 \\     46 \end{array} $
601 3 5 7 9	83 47 44 47 61	88 - 47 50 - 51 62	12 14 16 610	$     \begin{array}{r}       43 \\       18 \\       20     \end{array} $ $     \begin{array}{r}       18 \\       12     \end{array} $	$ \begin{array}{rrr}  & -34 \\  & 19 \\  & -22 \\  & -18 \\  & -10 \end{array} $
701 2	20 30	16 25	3 5 820	16 58	- 17 - 55
800 2 4 6 8 10	77 67 18 37 50 37	81 66 18 36 50 45	4 6 8 10 12 14	56 60 36 38 31 19	57 - 59 34 - 37 28 - 16
901 2 3 4 5 6	< 12 23 20 < 16 16 17	- 1 - 18 - 16 - 9 - 14 18	1010 3 1220 4 6	< 17 16 48 45 44	- 8 22 49 - 50 51
1001 3	50 33	54 33	8 10 12	29 25 25	$   \begin{array}{r}     -32 \\     34 \\     -28   \end{array} $

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hkl	$F_{\mathrm{o}}$	$oldsymbol{F_{\mathbf{c}}}$	hkl	$F_{ m o}$	$F_{\mathrm{c}}$
1620 4 6	23 23 21	$-\begin{array}{c} -21 \\ 23 \\ -25 \end{array}$	1680 10	19 16	- 15 - 18

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