detail. The results are given in Fig. 1. The disulfides I and II have a maximum at $371 \text{ m}\mu \ (\varepsilon_{\text{max}} = 58) \text{ and } 369 \text{ m}\mu \ (\varepsilon_{\text{max}} = 49),$ respectively. We think that this band is due to a transition from the antibonding $3p\pi$ -orbital to the antibonding σ -orbital of the disulfide group 3,4. This is in harmony with its almost complete disappearance in the dithiol III; the very small remaining absorption is certainly due to the precense of a trace of II formed by the unavoidable oxidation of III. We cannot at present give a precise interpretation of the absorption at the shorter wave-lengths. Remarkable is the strong absorption maximum at 268 m μ for the keto-compound (I). This cannot be due solely to the ketogroups. It may possibly be due to a transannular interaction of the 3-keto-group and the disulfide bridge (cf. Leonard et al. 5)

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Studies on the Zirconium-Aluminium and Hafnium-Aluminium Systems

LARS-ERIK EDSHAMMAR and STEN ANDERSSON

Institute of Inorganic and Physical Chemistry, University of Stockholm, Stockholm, Sweden

The following phases have been observed in the zirconium-aluminium system: Zr₃Al, Zr₂Al, Zr₅Al₂, Zr₄Al₃, ZrAl, Zr₂Al₃, ZrAl₂ and ZrAl₃. The crystal structures have been reported for Zr₃Al², Zr₅Al₃, ZrAl₂ and ZrAl₃.

In the present investigation, metals of high purity were used, viz. aluminium (99.99 %) and zirconium (iodide zirconium 99.9 %, oxygen 260 ppm, Foote Mineral Company). The alloys were prepared by melting in an electric arc furnace under an argon atmosphere and heat-treated in

sealed evacuated silica tubes. Tantalum foils protected the alloys from reacting with the silica.

The phase analyses were performed by means of X-ray powder patterns taken in a Guinier camera, potassium chloride being added to the specimens as an internal standard. Single crystal data were obtained in a Weissenberg camera with CuK radiation.

While the crystal structures of Zr_5Al_3 and $ZrAl_2$ were being studied at this Institute, Wilson et al.^{3,4} published the structures of these compounds. Thus, Zr_5Al_3 was reported to have the D8g type of structure while $ZrAl_2$ was found to be of the C14 type. The data given for $ZrAl_2$ are in perfect agreement with our results but this is, however, not the case with Zr_5Al_3 .

Single crystals of Zr₅Al₅ could be obtained from a melt of the approximate composition ZrAl_{0.50}. The single crystal and powder data showed the structure to be tetragonal with the following unit cell axes:

$$a = 11.049 \text{ Å}, \qquad c = 5.396 \text{ Å}.$$

The Patterson projections along[001] and [100] were calculated and the following structure was derived:

Unit cell content: 4 Zr₅Al₈. Space-group: I4/mcm (No. 140)

					æ	\boldsymbol{y}	z
16	\mathbf{Zr}	$_{ m in}$	16	(k)	0.077	0.218	0
4	\mathbf{Zr}	in	4	(b)	0	1/2	1/4
8	$\mathbf{A}\mathbf{I}$	in	8	(h)	0.16	$0.ar{6}6$	0
4	\mathbf{Al}	$_{ m in}$	4	(a)	0	0	1/4

This structure of the D8m type is represented by W₅Si₂ (Aronsson 7). It may be that the occurrence of the D8s and the $\mathbf{D}8_m$ structures depends on the heattreatment of the sample. It has, however, been suggested that the Me₅X₃ phases with the D8s type of structure are stabilized by the prescence of impurities such as boron, carbon, nitrogen and oxygen 8-10. This interesting hypothesis seems to be supported by the observation that a mixture of zirconium, aluminium and aluminium oxide with a composition close to the formula Zr₁₀Al₆O upon melting gave a product with a pure D8s structure. The powder pattern was found to give the following parameters for the hexagonal unit cell:

$$a = 8.177 \text{ Å}, \qquad c = 5.674 \text{ Å}.$$

The axes reported by Wilson et al.³ for the **D**8s phase are 8.184 and 5.702 Å, respectively.

The powder pattern of Zr₄Al₂ was obtained from a sample annealed at 850°C for several days. It was possible to index it it with the following hexagonal unit cell dimensions:

$$a = 5.430 \text{ Å}, \qquad c = 5.389 \text{ Å}.$$

The observed density, 5.32 g cm⁻³, is in good agreement with the value, 5.37 g cm⁻³, calculated for a cell content of one formula unit Zr_4Al_3 . The intensity data suggest that the structure contains chains of zirconium atoms $(Zr-Zr=\frac{1}{2}c)$ similar to those present in the Zr_4Al_3 phase.

Further studies on these structures are in

progress.

In connection with the investigations on the zirconium-aluminium system, the aluminium-rich region of the hafnium-aluminium system has also been studied. Thus, three compounds, viz. Hf₂Al₃, HfAl₂ and HfAl₃, have been prepared so far. The X-ray powder patterns demonstrate clearly that they are all isomorphous with the corresponding phases in the zirconium-aluminium system.

HfAl, has a hexagonal C14 type of structure with the unit cell axes:

$$a = 5.288 \text{ Å}, \qquad c = 8.739 \text{ Å}$$

HfAl₃ has a tetragonal D0₂₃ type of structure with the unit cell axes:

$$a = 4.010 \text{ Å}, \qquad c = 17.310 \text{ Å}$$

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Ozonolysis of p-Benzoquinone. III. Counter-current Ozonisation

ERLING BERNATEK and KRISTIAN VALEN

Universitetets Kjemiske Institutt, Blindern-Oslo, Norway

On two previous occasions 1,8 we have reported on the ozonolysis of p-benzo-quinone and discussed some of the possible reaction paths. Anomalous ozonolysis and/or rearrangement of normal ozonides occurred to a large extent. This was to be expected since in quinone two carbonyl groups are conjugated with the double bonds attacked by ozone.

The experimental procedure employed — being a batch procedure — had the drawback that the primary reaction products were in contact with ozone for a comparably long time. Consequently some of the isolated products could have been formed by further oxidation. In order to avoid this and especially to investigate closer the formation of a monozonide and its decomposition products, we adopted a counter-current procedure similar to that of Pummerer and Richtzenhain 3 which allowed a brief contact between the quinone solution and ozone, rapidly followed by hydrolysis.

Previous experiments with the batch procedure has shown that even if quinone was present in excess to the amount of ozone required for mono-ozonolysis only minor amounts of reaction products originating from cleavage of one double bond were found. It thus seems that secondary reactions proceed about as fast as the initial ozonolysis.

In Table 1 is summarised some of the relationships between the amount of applied ozone per mole of quinone, % reacted quinone and consumed ozone as found by the counter-current method. Finally the theoretically possible amount of monozonide (or products from mono-ozonolysis) in moles per mole of quinone is calculated.

When about one mole of ozone is applied per mole of quinone 65 % of the latter is attacked. A substantial part of the quinone must therefore have consumed two molecules of ozone. From the last column it is clear that mono-ozonolysis occurs in increasing amounts with decreasing ozone-quinone ratio.